



Preparation of a NaFePO_4 Cathode Material via Electrochemical Sodiation of FePO_4 Layers on Al Substrates

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Abstract. In this research, a cyclic voltammetry (CV) method was applied to intercalate Na^+ into an FePO_4/Al substrate to produce $\text{NaFePO}_4/\text{Al}$ as a potential cathode material. The sodiation was conducted directly to FePO_4 instead of applying delithiation to LiFePO_4 followed by sodiation, as was done in previous research. CV was conducted within a potential window of 2.0–4.0 V using a scan rate of 0.05 mVs^{-1} . The result was compared to $\text{LiFePO}_4/\text{Al}$ treated with a similar method. The various scan rate was then applied to understand its effect on the electrochemical activity recorded in the voltammogram and its impedance profile. The results show that the CV product of FePO_4/Al (NFP(A)) was crystallized in an orthorhombic olivine NaFePO_4 , as a result of Le Bail refinement. Orthorhombic $\text{Na}_{0.7}\text{FePO}_4$, trigonal FePO_4 , and monoclinic FePO_4 were presented as secondary phases. Meanwhile, the CV product of $\text{LiFePO}_4/\text{Al}$ (NFP(B)) was also crystallized in olivine NaFePO_4 and possessed secondary phases similar to NFP(A) with an additional Fe_2O_3 phase. NFP(A) showed two significant peaks at 2.442 V and 3.534 V, confirming sodiation/de-sodiation and $\text{Fe}^{3+}/\text{Fe}^{2+}$ activity, respectively. Meanwhile, NFP(B) showed two peaks at 3.183 V and 3.04 V, corresponding to de-lithiation and sodiation, respectively. The Nyquist plots of both materials show a similar profile, with the impedance value of NFP(A) being lower than that of NFP(B). This confirms that the CV treatment of FePO_4/Al is more facile than the treatment of the LiFePO_4 layer, while also producing a cathode with higher electrical conductivity. Scan rate reduction to 0.04 mVs^{-1} produced a much lower impedance value, confirming higher electrical conductivity.

Keywords: Electrochemical Na insertion; Electrical properties; Sodium ion-battery; Sodium iron phosphate

1. Introduction

As important components in Lithium Ion-Batteries (LIBs), positive electrodes or cathodes have been studied for a few decades, with LiFePO_4 (LFP) attracting the most attention due to its favorable kinetics in the lithium intercalation/de-intercalation process (Li et al., 2015; Sofyan et al., 2016; Yang et al., 2013), the ease of controlling its size and shape (Boesenberg et al., 2013), its low cost, its high thermal safety, its good reversible capacity, its long cycle ability, and the fact that it is environmentally friendly (Popović, 2011). However, sustainable lithium supply is now a concern because lithium is usually

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only mined from limited deposits of Li_2CO_3 and LiOH (Kim et al., 2019). Sodium-Ion Batteries (SIBs) might be a good alternative for LIBs due to their high natural abundance and uniform geographic distribution on earth (Li et al., 2015; Wang et al., 2016).

Iron-based phosphates have been investigated as SIB cathodes, such as in the molecular forms of $\text{Na}_2\text{FeP}_2\text{O}_7$ (Kim et al., 2013), $\text{Na}_2\text{FePO}_4\text{F}$ (Ellis et al., 2007), $\text{Na}_4\text{Fe}_3(\text{PO}_4)_2(\text{P}_2\text{O}_7)$ (Kim et al., 2012), NaFePO_4 (Kim et al., 2015), non-crystalline iron phosphate, FePO_4 (Liu et al., 2012), composites of amorphous FePO_4 nanospheres and carbon (Fang et al., 2014), and composites of FePO_4 nanospheres and graphene (Fan et al., 2014). Even though amorphous FePO_4 is easy to synthesize, olivine NaFePO_4 is preferable, especially for recently studied aqueous SIBs that require electrode materials to have operating potentials within the electrochemical stability window of water, in which H_2 and O_2 evolution should not occur (Jeong et al., 2019). The potential window should be between -0.83 V and 1.23 V vs. SHE at 25°C and 1 atm (Castellan, 1983). Meanwhile, maricite NaFePO_4 is not active electrochemically within 0 – 4.5 V due to cavities that trap Na ions and are not connected by pathways (Prosini et al., 2014). Further treatment to transform the maricite form into amorphous NaFePO_4 provides electrochemical activity within 1.5 – 4.5 V (Xiong et al., 2019). However, the potential window is above the potential window of aqueous SIB, which is in the range of -0.5 – 0.85 V (vs. Ag/AgCl) in 1 M aqueous NaClO_4 solution (Jeong et al., 2019).

NaFePO_4 can be conventionally synthesized through solid-state reactions using $\text{FeC}_2\text{O}_4 \cdot 2\text{H}_2\text{O}$, Na_2CO_3 , and $\text{NH}_4\text{H}_2\text{PO}_4$ as precursors, followed by calcination at 600°C for 10 h under an argon gas atmosphere (Kim et al., 2015). The maricite structure dominates the product. However, within the maricite structure, the diffusion channels of sodium ion transport are blocked (Heubner et al., 2017). Meanwhile, direct chemical synthesis appears to be unsuccessful for olivine NaFePO_4 preparation. Some researchers have produced olivine structures through ion-exchange methods using an organic-based electrochemical insertion into delithiated- FePO_4 (Oh et al., 2012; Avdeev et al., 2013; Galceran et al., 2014). This method requires two steps: de-lithiation and sodium insertion into the delithiated- FePO_4 in a new cell with sodium metal as the anode (Tang et al., 2016). The cyclic voltammetry profile of FePO_4 with LiClO_4 shows a single oxidation peak at 3.7 V and a single reduction peak at 2.85 V indicating de-lithiation/lithiation process of LiFePO_4 (Hansen et al., 2016). Meanwhile, when the NaClO_4 electrolyte was used, the peaks were at 3.65 V and 3.4 V, indicating de-lithiation of LiFePO_4 and then Na^+ insertion into the de-lithiated LiFePO_4 . The diffusion of Na^+ into FePO_4 is slower than that of Li^+ (Heubner et al., 2016). This means that lithiation competes the Na^+ diffusion easily.

Therefore, in this research, the Na^+ intercalation was conducted directly to the FePO_4 layer without passing through a de-lithiation step. The FePO_4 was cast on an Al substrate to produce FePO_4/Al which was then used as a working electrode. Cyclic voltammetry mode was applied within 2.0 – 4.0 V of potential windows with a scan rate of 0.05 mVs^{-1} . A similar treatment was also conducted to $\text{LiFePO}_4/\text{Al}$ for comparison. In addition, the effect of varying the scan rate effect was checked to determine the electrical properties of the materials.

2. Materials and Methods

2.1. Materials

The $\text{FePO}_4 \cdot 2\text{H}_2\text{O}$ (Sigma Aldrich, Singapore) powder was calcined at 700°C under argon gas (PT Samator Gas, Indonesia) flow before use. The calcined FePO_4 powder was then mixed with acetylene black (PT KGC, Indonesia) and polyvinylidene fluoride (PT KGC, Indonesia), with mass ratios of 70% , 20% , and 10% , respectively. The mixture was

dispersed in N-methyl-2-pyrrolidone (NMP) (PT KGC Indonesia), and then stirred until homogeneous. The slurry was cast on Al foil and dried at 60°C for 2 h in a vacuum oven. A similar treatment was also applied to the LiFePO₄ powder (MTI Corp., China) to produce LiFePO₄/Al.

2.2. The Electrochemical Sodiation

Electrochemical sodiation was conducted in a three-neck flask containing 50 mL of 1 M NaClO₄ electrolyte in propylene carbonate (analytical grade, Merck, Indonesia). The FePO₄/Al was used as the working electrode with Na metal (for synthesis, Merck, Indonesia) as the counter electrode. The reference electrode was Na metal. The electrodes were connected to a Corrtest CS150 electrochemical workstation (Corrtest, China) with a cyclic voltammetry mode of 2.0–4.0 V potential windows. The electrochemical cell was washed by allowing argon to flow for 5 min before operation.

After cyclic voltammetry, the sodiated FePO₄ (NFP(A)) was detached from the Al substrate. The NFP(A) powder was then heated to 470°C for 1 h under argon gas flow to remove the remaining NaClO₄ without destroying the olivine NaFePO₄ (Avdeev et al., 2013; US EPA, 2020).

2.3. Material Characterization

The synthesized-NaFePO₄ was characterized using X-ray Diffraction (XRD) (X-Pert PANalytical) equipped with RIETICA software for Le Bail refinement to identify the existing phases present and to predict the crystal structure and its cell parameters. Meanwhile, Scanning Electron Microscope/ Energy Dispersive X-ray (SEM/EDX) (JEOL JSM6510) analysis was conducted to understand the powder surface morphology. Electrical conductivity was investigated through impedance measurement with an Inductance-Capacitance-Resistance (LCR) meter (EUCOL U2826) by applying 0.2 V across a frequency range of 20 Hz to 5 MHz.

2.4. Electrochemical Analysis

For impedance measurement, a half-cell of NaFePO₄/Al was developed by applying a drop of NaClO₄ solution to the NaFePO₄/Al, followed by attaching the Al foil to cover the surface. This step was conducted within a glove box under an argon atmosphere. The half-cell was then vacuum sealed with an electric hot-press sealer. Both sides of the half-cell were attached with silver wire glued with silver paste. The NaFePO₄/Al side was connected to the positive pole, and the Al side was connected to the negative pole. A Nyquist plot of the measured-impedance Z , and the imaginary impedance Z'' , was fitted using the ZView software (included in the Corrtest Electrochemical Workstation) to get the resistance value R . ZView fitting was used to determine R_{total} (ohm), the resistance value when Z'' is set to zero. Meanwhile, θ is the angle (radian), and Cdl is the double layer capacitance (Farad). The calculated R_{total} was then used to calculate the electrical conductivity σ (Scm⁻¹) (Rahmawati et al., 2019).

3. Results and Discussion

The calcined FePO₄·2H₂O shows a significant change in the XRD pattern compared to the initial FePO₄·2H₂O, as seen in Figure 1a. Peaks at 2θ values of 27.60° and 58.52° (Figure 1b) confirm the presence of FePO₄ as they comply with the standard trigonal FePO₄ diffraction peaks (ICSD#412736). Meanwhile, peaks at 2θ values of 17.00°, 28.79°, 41.51°, 48.44°, and 53.00° comply with the peaks seen in the standard monoclinic FePO₄ diffraction pattern (ICSD#281079). Heat treatment also changed the surface morphology to larger particles with cleaned surfaces (Figure 1 insertion). EDX analysis found Fe, P, and O mass

contents of 44.11%, 18.37%, and 37.53%, respectively. Le Bail refinement found a trigonal FePO_4 P3121 that was crystallized within a volume cell of 269.912 \AA^3 with the following lattice parameters: $a = 5.20(2) \text{ \AA}$, $b = 5.20(2) \text{ \AA}$, $c = 11.44(4) \text{ \AA}$. A monoclinic structure P121/n1 with the following cell parameters was also found: $a = 5.57(2) \text{ \AA}$, $b = 7.62(2) \text{ \AA}$, and $c = 8.25(3) \text{ \AA}$.

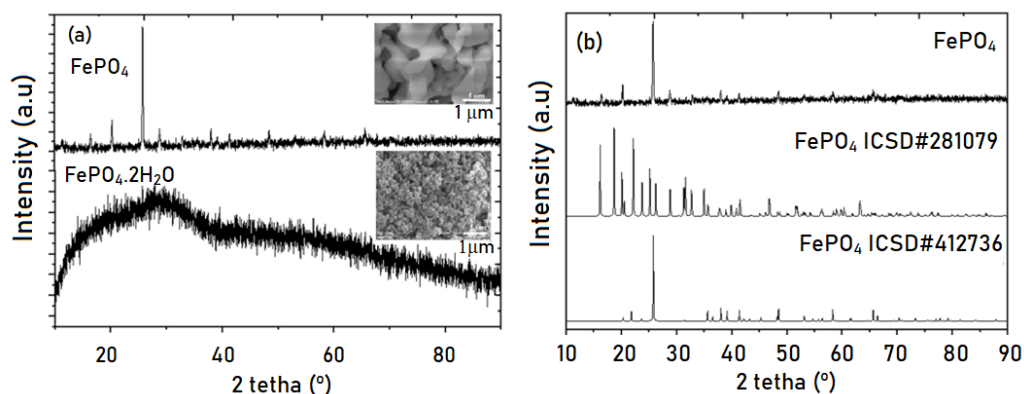
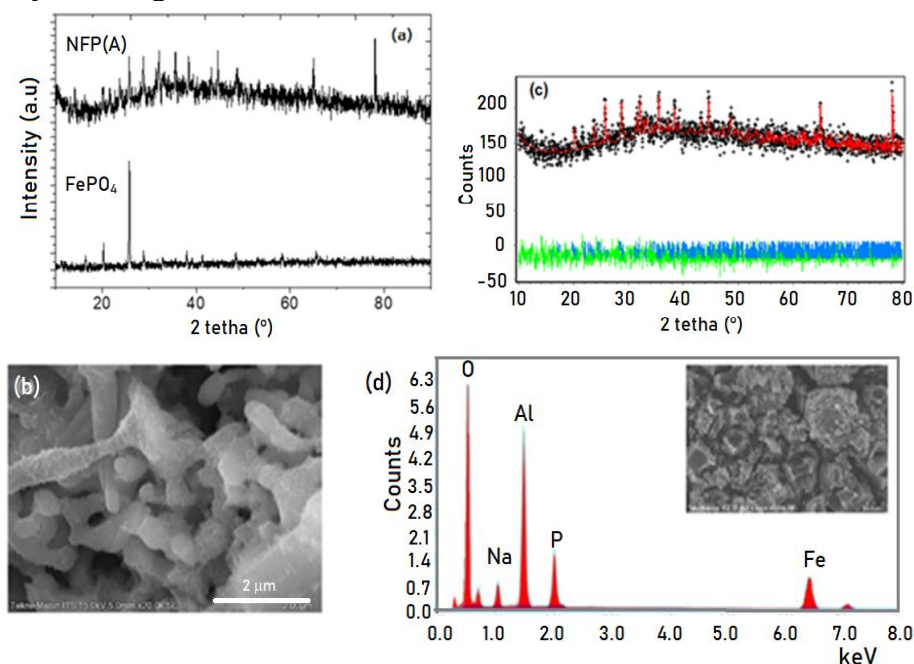


Figure 1 XRD patterns of: (a) $\text{FePO}_4 \cdot 2\text{H}_2\text{O}$ (bottom) and its calcined product FePO_4 (top); and (b) the XRD pattern of the experimental FePO_4 compared with standard FePO_4 XRD patterns (ICSD#412736 and ICSD #281079)

Electrochemical sodiation to the FePO_4 layer significantly changed the diffraction pattern, as presented in Figure 2, in which the peak intensity at $2\theta = 27.00^\circ$ decreased and some new peaks were revealed. Phase identification conducted using Le Bail refinement found that the new peaks fit well with orthorhombic NaFePO_4 and $\text{Na}_{0.7}\text{FePO}_4$, as described in the Le Bail plot in Figure 2c.



Elements	O (K)	Na (K)	Al (K)	P (K)	Fe (K)
Mass %	32.12	05.41	23.72	9.59	29.18
Error %	6.99	5.95	5.65	5.36	3.60
Atom %	50.77	9.88	22.23	7.83	13.21

Figure 2 (a) XRD patterns of FePO_4 and NFP(A) 0.05 mVs^{-1} ; (b) SEM images of NFP(A); (c) the Le Bail plot of NFP(A) by applying orthorhombic NaFePO_4 and orthorhombic $\text{Na}_{0.7}\text{FePO}_4$; and (d) the EDX result of NFP(A). The NFP(A) 0.05 mVs^{-1} is the NFP produced via electrochemical sodiation of FePO_4 with 0.05 mVs^{-1} of scan rate

The diffraction pattern is similar to that seen in literature on NFP production, specifically the high peaks at 2θ values of 27° , 29° , and 22° (Oh et al., 2012). Some small peaks at 15° , 20° , 32° , 36° , 37° , 40° , and 48° are also found (Oh et al., 2012), instead of peaks at 57.0° , and 67.5° (Tang et al., 2017). Le Bail refinement confirms the presence of orthorhombic olivine NaFePO₄ (Moreau et al., 2010) and orthorhombic Na_{0.7}FePO₄. The presence of orthorhombic Na_{0.7}FePO₄ was also detected in previous research (Casas-Cabanas et al., 2012), and was recently confirmed as an ordered Na_{2/3}FePO₄ composition (Boucher et al., 2014). SEM analysis confirms that electrochemical sodiation changes the particle shape from spherical grains (Figure 1a) into connected grains, as described in Figure 2b. EDX analysis recorded the presence of 9.88% atoms with 5.95% of errors, which results in a calculated $9.88 \pm 0.294\%$ atoms of Na (Ametek, 2020) (Figure 2d) inserted into FePO₄. The presence of Na inserted into FePO₄ was also proven by the XRD data, which showed that Na could be found in either orthorhombic NaFePO₄ or orthorhombic Na_{0.7}FePO₄. Al was also detected by EDX analysis (Figure 1d) as the FePO₄ film was cast on the Al substrate. The error percentage values were calculated by the TEAM-TM EDS software as a full calculation of analytical uncertainty and estimation of error propagation (Ametex, 2020).

The diffraction pattern of NFP(B) (Figure 3a) along with the Le Bail plot (Figure 3b) found that the diffraction pattern fit well with Na_{0.7}FePO₄ and NaFePO₄, however monoclinic FePO₄ was also found, and the Fe₂O₃ phase also existed. In fact, the Fe₂O₃ phase was also found in previous research on the synthesis of NaFePO₄ through the Pechini method (Sun et al., 2012). Le Bail refinement of XRD data found that cell parameters increased from LiFePO₄ ($a = 10.326(1) \text{ \AA}$, $b = 6.0082(7) \text{ \AA}$, $c = 4.6878(6) \text{ \AA}$, and cell volume = $290.83(6) \text{ \AA}^3$) to NaFePO₄ ($a = 10.682 \text{ \AA}$, $b = 6.290 \text{ \AA}$, $c = 5.138 \text{ \AA}$, and cell volume = 345.31081 \AA^3).

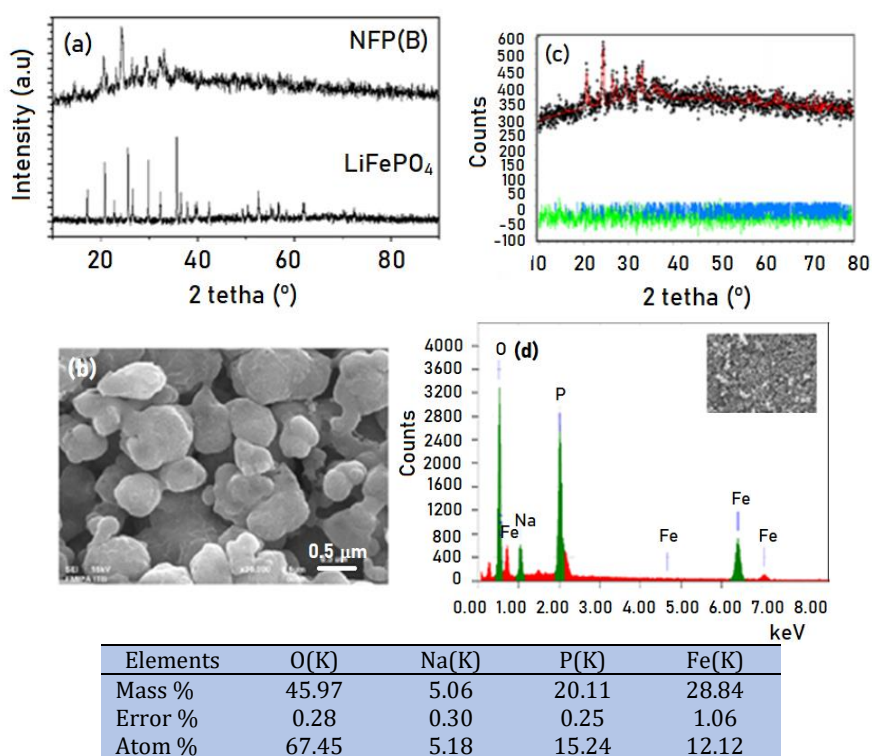


Figure 3 (a) XRD patterns of LiFePO₄ and its electrochemical sodiation result NFP(B); (b) its SEM image; (c) the Le Bail plot by applying orthorhombic NaFePO₄ and orthorhombic Na_{0.7}FePO₄; and (d) its EDX results

The $\sim 0.26\text{--}0.28$ Å increase along the a and b axes corresponds to the replacement of Li (90 pm) with Na (110 pm). Meanwhile, Moreau et al. (2010) found a similar finding in their preparation of NaFePO₄ from LiFePO₄ with a potentiodynamic (PITT) mode, where a 0.2 Å increase was observed for the b and c axes (Moreau et al., 2010).

Cyclic voltammetry at 2.0–4.0 V did not completely transform the intermediate phase of Na_{0.7}FePO₄ into NaFePO₄. The intermediate Na_{0.7}FePO₄ was successfully isolated in previous research to study Na insertion/extraction to a commercial carbon-coated LiFePO₄, in which further sodiation was chemically carried out by NO₂BF₄ as an oxidizing agent and NaI as a reducing agent (Casas-Cabanas et al., 2012). Previous research confirmed that the Na_{0.7}FePO₄ is in the orthorhombic *Pnma* intermediate phase. The XRD patterns in this study matched those found in previous research. Casas-Cabanas et al. (2012) found that the cell parameters increase within the b [010] and a [100] directions after Na insertion, instead of along the c direction, as observed in electrochemically prepared samples (Moreau et al., 2010). Both findings indicate the probability of cell parameters increasing along the a, b, and c axes due to Na insertion, as it was also found within this study that the c cell parameter increases from 4.6878(6) Å in LiFePO₄ to 5.138 Å in NFP(B). Elemental analysis confirmed Na insertion by recording a 5.18% Na presence within the NFP(B) (Figure 3d).

Typical Nyquist plots of NFP (A) and NFP (B) are also similar (Figure 4b). ZView fitting determined the electrical conductivity of the materials to be $7.69 \times 10^{-7} \text{ Scm}^{-1}$ and $6.10 \times 10^{-7} \text{ Scm}^{-1}$ for NFP(A)0.05 and NFP(B)0.05, respectively (Table 1). Both instances are categorized as electrode-bulk interface conduction, based on their respective capacitance values of $7.69 \times 10^{-7} \text{ F}$ and $6.10 \times 10^{-7} \text{ F}$, as a capacitance of 0.1 μF is considered the cut-off value for electrode-bulk interfaces (Martín et al., 2007; Apriyani et al., 2016; Rahmawati et al., 2021). The electrical conductivity is higher than today's leading cathode material, LiFePO₄, which has electrical conductivity as low as 10^{-9} Scm^{-1} (Amin et al., 2008), and $2 \times 10^{-8} \text{ Scm}^{-1}$ for Li₃V₂(PO₄)₃ (Yin et al., 2003) at room temperature.

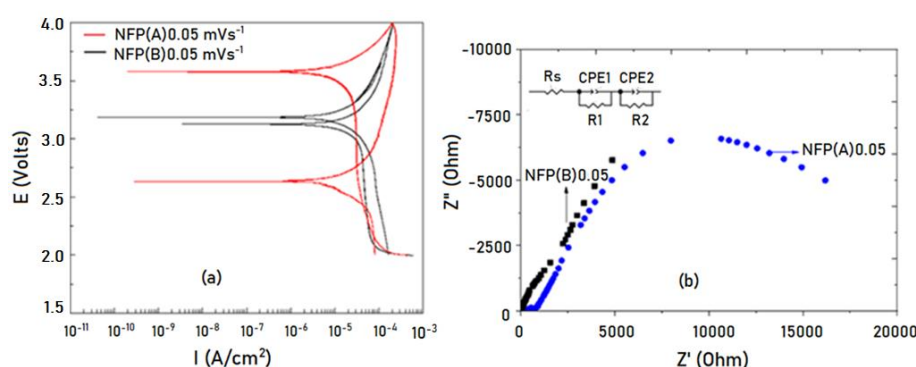


Figure 4 (a) CV curve of NFP(A)0.05 mV/s and NFP(B)0.05 mV/s; and (b) Nyquist plots of NFP(A) and NFP(B) along with their R-C network model

Table 1 Resistance and conductivity of Al | NaFePO₄ | NaClO₄ | Al for NFP(A) and NFP(B)

Material	R(Ω)	Capacitance (Farad)	Conductivity (Scm ⁻¹)	Chi square (χ ²) (Goodness of fit*)
NFP (A)0.04 mVs ⁻¹	1242	1.95×10^{-7}	5.84×10^{-6}	0.399
NFP (A)0.05 mVs ⁻¹	22251	7.69×10^{-7}	5.93×10^{-7}	0.019
NFP (A)0.2 mVs ⁻¹	227960	1.22×10^{-7}	2.27×10^{-8}	0.665
NFP (B)0.05 mVs ⁻¹	30590	2.69×10^{-6} 6.10×10^{-7}	1.75×10^{-7}	0.048
NFP(B)0.04 mVs ⁻¹	4309	6.51×10^{-7}	1.79×10^{-6}	0.149

*Significant level is in between 0–1 (STAT TREK, 2020)

The volume expansion after Na insertion to FePO₄ to form NaFePO₄ is 12.999% ($\Delta V = 35.086 \text{ \AA}^3$), which is 2.69 times higher than the volume change of 4.836% for Na_{0.7}FePO₄ to NaFePO₄ formation ($\Delta V = 14.748 \text{ \AA}^3$). This is in agreement with previous research by Fang et al. (2015), who found that the volume expansion from FePO₄ to Na_{2/3}FePO₄ is three times larger than from Na_{2/3}FePO₄ to NaFePO₄ (Fang et al., 2015). It is known that by increasing the scan rate 10–40 times, the rate constant of Na_{0.7}FePO₄ formation increases, and two peaks are revealed within the oxidation and reduction parts. In this research, the increase in scan rate up to 0.2 mVs⁻¹ also revealed two peaks at 2.48 V and 2.66 V (Figure 4), which is close to another result (Fang et al., 2017).

This research also applied a low scan rate of 0.04 mVs⁻¹, lower than the scan rate of 0.05 mVs⁻¹ applied in other studies (Fang et al., 2015; Moreau et al., 2010), to understand the CV type when the current density was reduced to $4.259 \times 10^{-8} \text{ Acm}^{-2}$. The peak split into two (Figure 5a), similar to NFP(A)0.05 mVs⁻¹; however, the second peak was revealed at a lower voltage of around 2.8 V. This probably indicates that one or both of Na⁺/Na activation or Fe³⁺/Fe²⁺ activation occurred and resulted in a higher electrical conductivity. Low current density seems to provide a low insertion/extraction rate which may produce a more regular structure with a low impedance value (Figure 5b). Further structural studies are required to deeply investigate this issue. In addition, low current density also revealed roughness peaks that probably came from multiple phases of the prepared NaFePO₄. Another study found that the presence of impurities in multi-walled carbon nanotubes plays an important role in catalyzing electrochemical reactions (Mustafa et al., 2018). Another study on metallic impurities within Double-Walls Carbon Nano Tubes (DWCNTs) also determined the effect of metals on the electrochemical response of the DWCNT electrode (Pumera and Iwai, 2009).

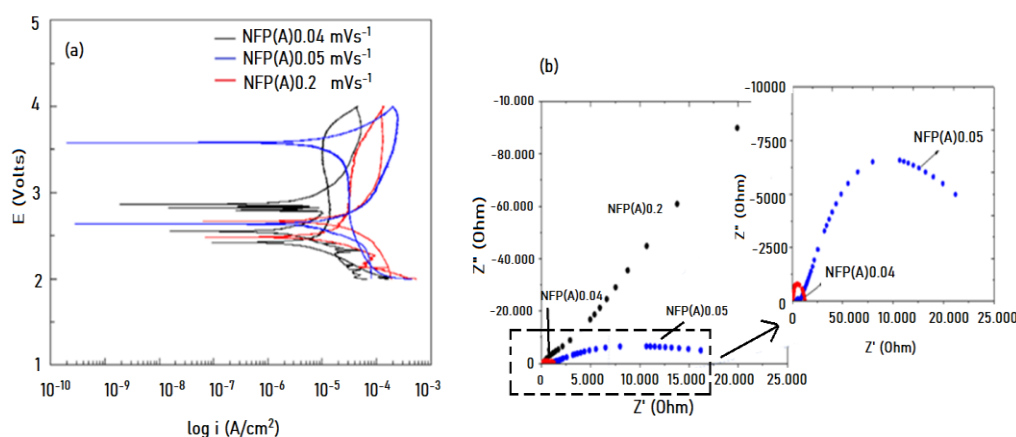


Figure 5 (a) Cyclic voltammogram of NFP(A) under different scan rates; and (b) their impedance plots

Nyquist plots in the range of 20 Hz–5 MHz (Figure 5b) show two semicircles without a spike line because spike lines are usually only revealed at very low frequencies below 20 Hz, as seen in the work of Zhu et al. (2013), which showed a spike line in a Nyquist plot of C-NaFePO₄ at recorded impedance frequencies as low as 10⁻³ Hz. The high-frequency semicircle is attributed to grains or particle-particle impedance or particle-to-current collector impedance (Illig et al., 2012; Zhu et al., 2013), while the middle frequency semicircle corresponds to charge transfer resistance (Illig et al., 2012). The semicircle of NFP(A)0.04 mVs⁻¹ falls into a Z' value of 1242 Ω (Table 1), and the value is comparable to the impedance of mixed carbon NaFePO₄ (C-NaFePO₄) with the semicircle falling to a Z'' value of around 1300 Ω (Zhu et al., 2013). This implies that NFP(A)0.04 mVs⁻¹ is sufficient

for an electrochemical full cell. A similar rate of 0.04 mVs^{-1} was also applied to NFP(B) and produced a similar semicircle to that of NFP(A) 0.04 mVs^{-1} with an impedance value of 4309Ω , which was a significant reduction from 30590Ω provided by NFP(B) 0.05 mVs^{-1} (Table 1). It would be interesting for future work to optimize the electrochemical scan rate to find the highest conductivity of the NFP cathode.

4. Conclusions

Electrochemical sodiation was successfully applied in cyclic voltammetry mode to insert Na ions into the FePO_4 layer directly. This finding offers a simpler electrochemical method to insert/extract Na^+ directly as an alternative to the two-step de-lithiation and sodiation method. Impedance analysis even shows a similar semicircle trend, with a lower impedance value of NaFePO_4 from FePO_4 , NFP(A), than the one prepared from the LiFePO_4 layer, NFP(B). This research also found that by reducing the scan rate to as low as 0.04 mVs^{-1} , the electrical conductivity increased for both NFP(A) and NFP (B).

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